

## OPTICAL PROPERTIES OF COMPOUNDS WITH SUBMICRON POINTS OBTAINED THROUGH $Ga_2S_3$ INTERCALATION WITH Cd

Racoveț O.<sup>1</sup>, Evtodiev I.<sup>1\*</sup>, Caraman Iu.<sup>2</sup>, Rotaru I.<sup>1</sup>, Lazăr G.<sup>2</sup>

Moldova State University, 60, Alexei Mateevici str., Chisinau, MD-2009, Republic of Moldova

“Vasile Alecsandri” University of Bacau, 157, Calea Marasesti, Bacau, 600115, Romania

\*e-mail: [ievtodiev@yahoo.com](mailto:ievtodiev@yahoo.com)

Luminescence and optical absorption spectra of  $Ga_2S_3$  single crystals were investigated at temperatures of 78 K and 293 K. Optical band gap is equal to 3.27 eV and 3.457 eV at 293 K and 78 K respectively. Luminescence spectrum of single crystal lamellas at temperature of 78K consists of three bands with peaks at 2.04 eV, 1.84 eV and 1.66 eV. Native structural defects form deep recombination and electronic capture levels localized within the  $Ga_2S_3$  band gap.

Keywords:  $Ga_2S_3$ , single crystals, intercalation, photoluminescence, absorption.

Au fost obținute spectrele de luminiscentă și absorbție ale monocristalelor  $Ga_2S_3$  la temperaturi de 78 K și 293 K. Banda optică interzisă are mărimea de 3,27 eV la 293K și 3,457 eV la 78 K. Spectrul de luminiscentă al peliculelor monocristalului pentru temperatura de 78 K este format din trei benzi cu valorile maxime de 2.04 eV, 1.84 eV și 1.66 eV. Defectele structurale proprii inițiază formarea nivelelor de recombinare și captare de electroni, localizate în interiorul benzii interzise.

Cuvinte-cheie:  $Ga_2S_3$ , monocristale, intercalare, fotoluminescență, absorbție.

### INTRODUCTION

Interest towards  $A_2^{III}B_3^{VI}$  semiconductors, which are still little studied, has increased with appearance of papers demonstrating that compounds of this group may have technical applications in various fields of optoelectronics, photonics and interferential optics [1 - 6].

$Ga_2S_3$  belongs to the class of materials with native structural defects  $Ga_2S_3$ , which are less investigated. High concentration of native structural defects determines the physical properties of this material, and primarily, the low electrical conductivity. Being a semiconductor with wide band gap,  $Ga_2S_3$  is characterized by strong photoconductivity in the blue-violet region of the spectrum [7, 8].  $\beta$ - $Ga_2S_3$  has wurtzite-type structure with cell parameters:  $a = 3,678\text{Å}$ ,  $c = 16,018\text{Å}$  [9, 10]. Due to specific structural bonds, single crystals of  $\beta$ - $Ga_2S_3$  can be easily split in the perpendicular direction to the  $C$  axis.  $Ga_2S_3$  is considered a prospective material for various technological applications: neutral layer heterojunctions [4, 5, 11], optical and optoelectronic applications [12, 13], photovoltaic structures [14, 15] and luminescent screens [16, 17]. In recent years, the attention of researchers is focused on

radiative properties of gallium sulphide doped with  $Mn^-$ ,  $Ag^-$ ,  $Cu^-$ ,  $Ge$  și  $Sm$  [19-21]. These dopants form deep recombination levels and luminescence centers related to vacancy of metal sublattice.

Fundamental band edge absorption (AO) spectra and photoluminescence (PL) spectra of  $Ga_2S_3$  compound intercalated with Cd in vapor phase are analyzed in the paper.

### EXPERIMENTAL METHODS

PL and AO spectra were measured for plan-parallel plates of  $Ga_2S_3$  and  $Ga_2S_3$  intercalated with Cd in vapor phase. Optically homogeneous  $Ga_2S_3$  single crystal layers with 2...8 mm<sup>2</sup> area and thickness varied from ~50 μm to 1...2 mm, were obtained using  $I_2$  vapor transport at normal atmospheric pressure. As transport material, single crystals grown by Bridgman method from primary elements of Ga (4N) and S in stoichiometric proportions were used. Optically transparent plates of  $\beta$ - $Ga_2S_3$  with perfect surfaces were grown at temperature gradient of ~20° cm<sup>-1</sup>. Single crystal plates of  $\beta$ - $Ga_2S_3$  were intercalated with Cd from vapor phase by means of heat treatment at  $T = 500\text{ K}$  for 6 h.

Photoluminescence at 78 K and 293 K was registered for single crystal layers of  $Ga_2S_3$  with the optical axis ( $C_6$ ) perpendicular to the sample surface. Cd excess was evaporated from the  $Ga_2S_3$  plate surface by heat treatment in vacuum at temperatures of 400...420 °C for 30 min.

The spectral characteristics of AO and PL spectra were registered using a spectrophotometer assembled on MDR-2 monochromator with diffraction grating (1200 mm<sup>-1</sup> and 600 mm<sup>-1</sup>). As a light source for optical measurements (transmission and reflection) at temperature of 293 K, a lamp with W filament was used. Photoluminescence was excited with He - N<sub>2</sub> laser radiation ( $\lambda = 337,4$  nm). Excitation radiation density was varied using neutral density filters (thin layers of Pt on amorphous quartz supports).

## EXPERIMENTAL RESULTS

Absorption spectra of  $\beta - Ga_2S_3$  single crystals with hexagonal crystalline structure are shown in Figure 1 at 78 K (curve 1) and 293 K (curve 2). Two pronounced sectors with slopes of 3,15...3,20 eV and 3,33...3,35 eV are manifested on the line of  $\alpha(h\nu)$  dependence at  $T=293$  K. The second sector probably corresponds to the direct optical transitions between the valence band (VB) top and conduction band (CB) bottom. Fundamental band edge absorption in  $Ga_2S_3$  crystals at 293 K is located in the energy range of 3,32...3,36 eV. Extrapolating linear segment of the  $f(h\nu)$  dependence at  $\alpha \rightarrow 0$ , it was appreciated the width of band gap of  $\beta - Ga_2S_3$  crystals equal to 3,27 eV. Relatively high values of absorption coefficient in the region of 3,20...3,33 eV are caused by the interaction of photon-electron-phonon type.

At low temperatures (Figure 1, curve 1), two narrow absorption lines are manifested at 3,48 eV and 3,42 eV. Rapid decrease of these bands intensity with increasing temperature indicates the excitonic nature of these lines. Considering the excitons in  $\beta - Ga_2S_3$  crystals with structural defects as having large radii (Vanier Mott excitons), the binding

energy of the electron-hole pair is estimated to be equal to 72 meV, width of band gap is equal to 3,475 eV at  $T=78$  K.

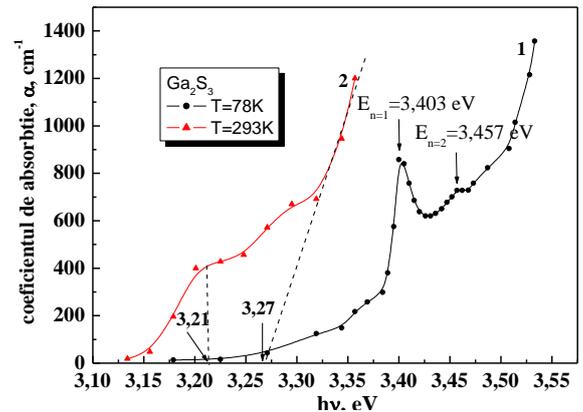


Fig. 1.  $\beta - Ga_2S_3$  single crystals absorption spectra at 78 K (curve 1) and 293 K (curve 2).

Photoluminescence spectrum of  $\beta - Ga_2S_3$  crystals at  $T=78$  K (Fig. 2) consists of a dominant intensity band at 1,873 eV and a low intensity one with maximum at 2,84 eV. Since  $Ga_2S_3$  crystals are of  $n$ -type, energy of levels corresponding to these PL bands can be equal to 0,43 eV and 1,40 eV above the valence band maximum.

Relatively large width of the dominant PL band and the lack of vibrational structure indicates a strong phonon emission. PL band half-width in the temperature range of 80 ÷ 293 K weakly depends on temperature (increases by about 30 meV), which is in good correlation with the Debay temperature ( $\theta \approx 280$  K).

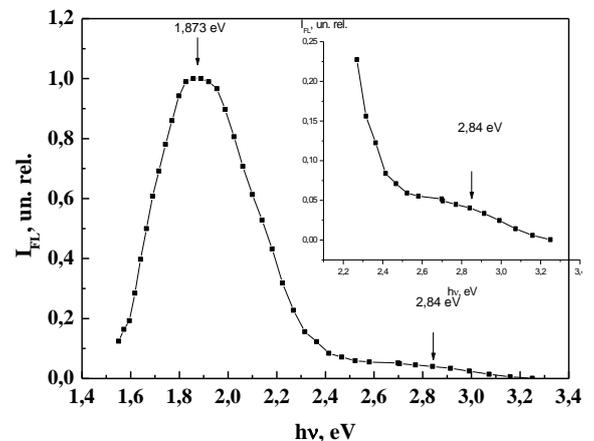


Fig. 2. Photoluminescence spectra of  $Ga_2S_3$  single crystal at 78 K.

If we admit that the luminescence is caused by radiative transitions of electrons from the conduction band to deep acceptor

levels, then the energies of these two levels of recombination (electron-hole) are found at 0,72 eV, and, respectively, at 1,62 eV above the valence band maximum.

Currently, there is no well-argued theory regarding the energy states of intrinsic and dopant electros. But analyzing the obtained results for two limiting cases, the strongly doped semiconductors [4, 5] and amorphous semiconductors [6], we can admit that the electronic bands on the edge of Brillouin band are easily curved, entering the band gap. Thus, optical transitions can occur between the *VB* and *CB* tails. Relatively high absorption coefficient in the middle slope can be explained by the presence of these absorption mechanisms.

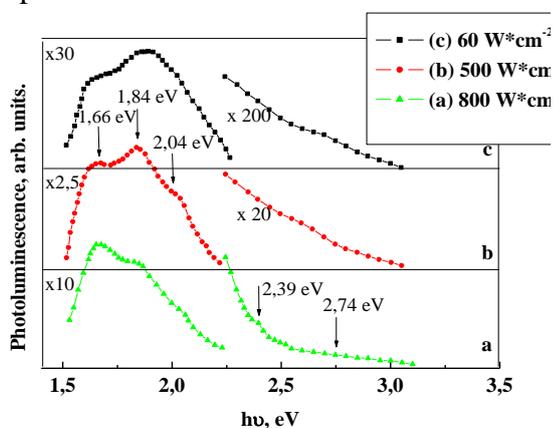


Fig. 3. Photoluminescence spectra of  $Ga_2S_3$  single crystal layers ( $d=58 \mu m$ ) at temperature 78 K for various excitation densities.

Further, we analyze the the photoluminescent properties of  $Ga_2S_3$  single crystal plates with thickness of 58  $\mu m$ . *PL* spectra at three excitation densities (800  $W/cm^2$ , 500  $W/cm^2$  and  $\text{\textasciitilde}$  60  $W/cm^2$ ) of  $Ga_2S_3$  single crystal plates at 78K are shown in Fig. 3. As we can see, *PL* spectra cover the wide range of energies from 1,5 eV up to 3,1 eV. *PL* intensity slowly increases in the range of absorption band edge, which, as noted above, may be determined by electronic transitions between the energy bands tails. The low intensity of photoluminescence is probably caused by high probability of non-radiative transitions between these states.

The intensive luminescence begins at energies less than 2,5 eV. Starting with the energy of 2,5 eV, *PL* increases rapidly irrespective of excited beam density and

reaches the maximum value at 1,84...1,95 eV. The *PL* band in this spectral region is due to overlapping of three *PL* bands with maxima at 2.04 eV, 1.84 and 1,67...1,65 eV. As we can see, comparing the curves a, b and c (Fig. 3), the intensity of this band depends ambiguously on the excitation beam intensity. Peculiarities at 2,39 eV and 2,74 eV (Fig. 3) are determined by the presence of two acceptor levels localized at 0,92 eV, and, 0,47 eV respectively above the valence band edge [7].

*PL* spectra of  $Ga_2S_3$  single crystals subjected to the heat treatment at 480  $^{\circ}C$  for 6 h in *Cd* vapors are shown in Fig. 4, a. Energy of *PL* bands, as well as probable interpretation of the *PL* spectrum structure are presented in the Table.

Particles energy in absorption and luminescent spectra of  $Ga_2S_3$  and  $Ga_2S_3:Cd$  single crystals.

Compound	Optical band gap $E_g$ , eV		Luminescence, arb. unit.					
	293 K	78 K	B <sub>1</sub>	B <sub>2</sub>	B <sub>3</sub>	B <sub>4</sub>	B <sub>5</sub>	
$Ga_2S_3$	3.27	3.457	2.74	2.39	2.04	1.84	1.66	
$Ga_2S_3$ tratat în vapori de Cd			A <sub>1</sub>	A <sub>2</sub>	A <sub>3</sub>	A <sub>4</sub>	A <sub>5</sub>	A <sub>6</sub>
			2.92	2.75	1.90	1.76	1.64	1.38

The *PL* integral intensity of the dominant band ( $h\nu \approx 1,83eV$ ) decreases by more than 5 times in the temperature range from 78 K up to 293 K (room temperature). The temperature quenching of the dominant *PL* band with a maximum in the region of 1,64...1,76 eV is shown in Fig. 4, b. As one can see, the *PL* intensity (*L*) of  $Ga_2S_3$  crystals intercalated with *Cd* is well described by the formula:

$$L = L(0) \left( 1 + \exp \frac{\Delta E}{kT} \right) \quad (1)$$

where  $\Delta E$  is the energy of temperature quenching of photoluminescence,  $L(0)$  is *PL* intensity at  $T=0K$ ,  $K$  is the Stefan-Boltzmann constant.

The *PL* intensity decreases by exponential law with the factor of 60 meV in the temperature range of 79...293 K, and this quenching is stronger at higher temperatures, the exponent factor being 100 meV. Thus, in  $Ga_2S_3$  crystals intercalated with *Cd*, the thermal activation energy for orange band is 60 meV and 100 meV.

The *PL* spectra of  $Ga_2S_3$  crystals at excitation densities from 60 W/cm<sup>2</sup> to 800 W/cm<sup>2</sup> are presented in Fig. 5. As one can see, the *L(W)* dependence may be described by a power law of type:

$$L \propto W^p \quad (2)$$

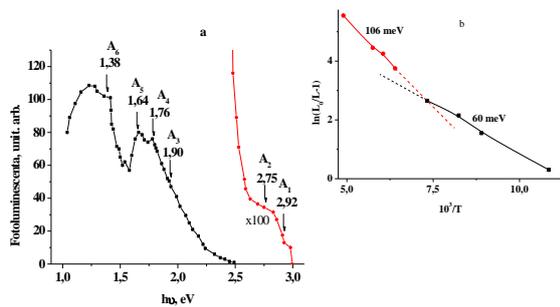


Fig. 4. Photoluminescence spectra of  $Ga_2S_3$  single crystals intercalated with *Cd* (a) and temperature quenching of the dominant *PL* band with a maximum in the region of 1,64...176 eV (b).

The *p* index is a photoluminescence kinetics characteristic, which can be found from Fig. 5. This parameter is equal to 0.9 and 1,0 for 2,04 eV and 1.84 eV bands respectively, that indicates that these bands are due to monomolecular kinetic processes. The *p* parameter is equal to 1,4 for 1,65 eV band and this band is of recombination nature.

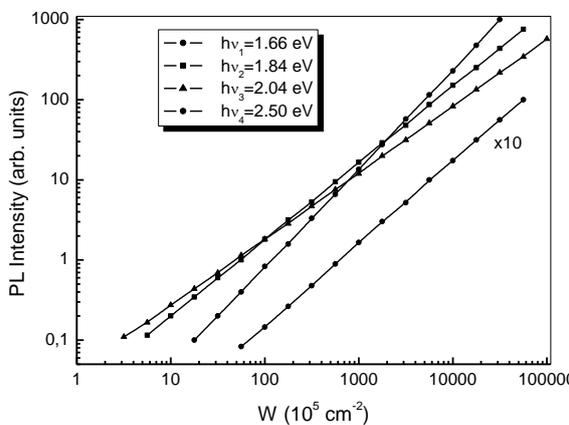


Fig. 5. The dependence of *PL* intensity, *L*, on excitation density power, *W*, for single crystal layers of  $Ga_2S_3$ .

From the analysis of the absorption and photoluminescence spectra at temperature 78 K and 293 K, as well as temperature quenching, the diagrams of possible radiative energy transition in  $Ga_2S_3$  and  $Ga_2S_3$  crystals intercalated with *Cd* were elaborated (Fig. 6).

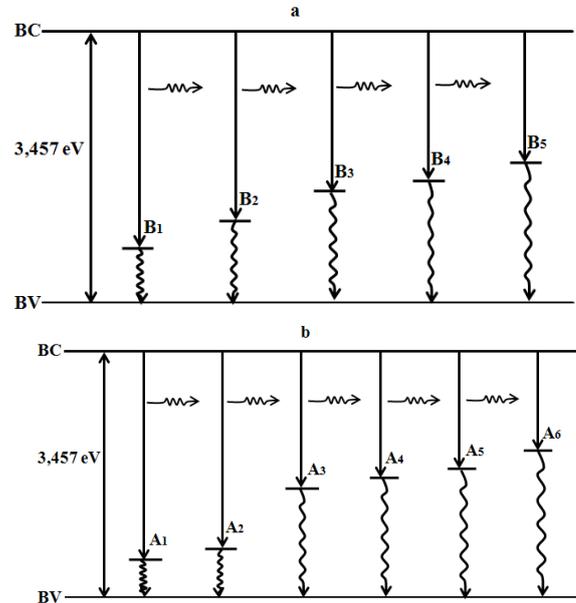


Fig. 6. Diagram of recombination levels in  $Ga_2S_3$  (a) and  $Ga_2S_3$  crystals intercalated with *Cd* (b) at temperature of 78K.

### CONCLUSIONS

Heat treatment of single crystals in *Cd* vapors at temperature of 480 °C changes the diagram of levels of recombination luminescence. *Cd* atoms intercalated in  $Ga_2S_3$  form recombination deep levels with energy of 0,72 eV and 1,62 eV.

The diagram of radiative levels within the band gap of  $Ga_2S_3$  compound, formed as a result of heat treatment in *Cd* vapors at temperature 480 °C, was elaborated.

It was established that direct optical transitions in single crystal layers of  $\beta-Ga_2S_3$  take place both at the temperature of 78 K and 293 K. The band gap energy is equal to 3,27 eV at temperature of 293 K, and 3,457 eV at 78 K.

### REFERENCES

1. Webster J. G. (Ed.), The Wiley Encyclopedia of Electrical and Electronics

Engineering, Electronic Materials. John Wiley & Sons, New York, 2001. p. 17616.

2. Afzaal M., O'Brien P. J. Recent developments in II–VI and III–VI semiconductors and their applications in solar cells. *J. Mater. Chem.*, 2006, 16, p. 1597-1602.

3. Caraman M., Cuculescu E., Evtodiev I. Bistable optic modulators made from monocrystalline layers GaSe (Cu). *J. Optoelect. Adv. Mater.*, 2005, 7 (2), p. 805-810.

4. Matore H. F. Defect Electronics in Semiconductors. John Wiley & Sons, New York, 1971. P. 639.

5. Fistul V. I. Heavily Doped Semiconductors (Translated from the Russian by A. Tybulewicz). Plenum Press, New York, 1969. P. 418.

6. Mott N. F., Davis E. A. Electronic Processes in Noncrystalline Materials, 2<sup>nd</sup> ed., Clarendon Press, Oxford, 1979. P. 590.

7. Mushinski V. P., Caraman M. I. Photoelectrical and Luminescent Properties of Gallium and Indium Chalcogenides. Chisinau, Stiinta, 1975, p. 61 (in Russian).

8. Medvedev Z.S. Group III b Chalcogenides. Moscow, Nauka, 1968, p. 18-45 (in Russian).

9. Tomas A., Pardo M.P., Guittard M., Guymont M., Famery R. Determination des structures des formes  $\alpha$  et  $\beta$  de  $Ga_2S_3$  structural determination of  $\alpha$  and  $\beta$   $Ga_2S_3$ . *În: Mater. Res. Bull*, 1987, 22, p. 1549-1554.

10. Medvedeva S.A. (Ed.), Physico-chemical Characteristics of Semiconductor Compounds. Nauka, Moscow, 1979. P. 339 (in Russian).

11. Yüzer H., Doğan H., Koroğlu J., Kocakuşak S. Analysis of sulfide layer on gallium arsenide using X-ray photoelectron spectroscopy. *Spectrochimica Acta Part B: Atomic Spectrosc.*, 2000, 55, p. 991-996.

12. Takebe H., Kitahawa R., Hewac D.W. Non-Toxic Sulfide Glasses and Thin Films for Optical Applications. *Journal of the Ceramic Society of Japan*, 2005, 113 (1), p. 37-43.

13. Caricato A.P., Fernández M., Leggieri G., Luches A., Martino M., Montagna M., Prudeniano F., Jha A. Chalcogenide glass thin film waveguides deposited by excimer laser ablation. *Applied Surface Science*, 2003, 208-209, p.632-637.

14. Weber A., Kotschau I., Shock H.W. Monitoring In–Ga interdiffusion during chalcopyrite formation in  $Ga_xS_y$ –(Cu,In) photovoltaic precursor layers. *Thin Solid Films*, 2007, 515, p. 6252-6255.

15. K. F. Abd El-Rahman. Charge conduction mechanisms and photovoltaic properties of n-( $Ga_2S_3$ - $Ga_2Se_3$ )/p-Si heterojunctions. *The European Physical Journal - Applied Physics*, 2007, 37 (2), p. 143-147.

16. Okamoto S., Tanaka K., Inoue Y. Blue emitting Sr 2 Ga 2 S 5 :Ce phosphor thin films grown by multisource deposition. *Appl. Phys. Lett.*, 2000, 76, p. 946-948.

17. Nakano, F.; Uekura, N.; Nakanishi, Y.; Hatanaka, Y.; Shimaoka, G. Preparation of  $CaGa_2S_4$ :Ce thin films for blue emitting thin-film EL device. *Appl. Surf. Sci.* 1997. 121/122, p. 160-162.

18. Springford M. The luminescence characteristics of some group III-VI compounds. *Proc. Phys. Soc.*, 2000, 82, 1020-1028.

19. Lee J.-S., Won Y.-H., Kim H.-N., Kim C.-D., Kim W.-T. Photoluminescence of  $Ga_2S_3$  and  $Ga_2S_3$ :Mn single crystals. *Solid State Commun.*, 1996, 97 (12), p. 1101-1104.

20. Aono T., Kase K. Photoemission of Ag, Cu, and Ge-doped  $\alpha$ - $Ga_2S_3$  crystals. *Solid State Commun.*, 1992, 83 (10), p. 749-752.

21. Georgobiani A. N., Tagiev B. G., Tagiev O. B., Ganbarova Kh. B. Photoluminescence of  $Ga_2S_3$ : $Sm^{2+}$  crystals. *Inorg. Mater.* 2008, 44 (6), p. 563-565.

22. Birkle G. V. B., Gavrilo F. F., Kitaev G. A. Anomalous thermal quenching of luminescence in ZnS:Cu phosphor crystals. *Sov. Phys. J.*, 1979, 22, p. 718-722.

23. Aono T., Kase K. Green photoemission of  $Ga_2S_3$  crystals. *Solid State Commun.*, 1992, 81 (4), p. 303-305.

Prezentat la redacție la 10 septembrie 2012